Surface Characterization of Thorium-Nickel Intermetallic Catalysts

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Electron spectroscopy for chemical analysis (ESCA), Auger electron spectroscopy (AES), and ion scattering spectroscopy (ISS) have been used to study a series thorium-nickel alloys (Th₇Ni₁, ThNi, ThNi₂, ThNi₅), both untreated and treated with different gases (O_2 , air, CO/H₂). ESCA results indicate that the surfaces of untreated alloys are Ni-poor with respect to the bulk. Treatment of these alloys with synthesis gas $(CO/H₂)$, $O₂$, or air results in decomposition of the intermetallic into Ni/ThO₂ or NiO/ThO₂. In all cases Ni was observed to segregate to the surface after reaction. The degree of Ni surface segregation is not affected by different gas treatments but varies for different alloys. The relationship between the relative Ni signal intensity and the bulk Ni content was examined and compared by ESCA, AES, and ISS which have different sampling depths. Catalysts with large surface concentrations of Ni generally show high activity for the methanation reaction. Comparison of the Ni surface concentration with catalytic activity of Th₇Ni₃ and ThNi₅ has shown that they do not correlate mathematically. This incompatibility can be explained by the presence of different active sites, a spongy model. or both.

INTRODUCTION

A new class of supported catalysts has been developed $(1-4)$ recently which possesses higher activity than conventional supported catalysts prepared by wet chemical techniques. The new catalysts are formed by reaction of synthesis gas (CO/ $H₂$) or $O₂$ with a binary intermetallic compound which generally consists of a rare earth or actinide element and a first-row transition metal. This conversion may be expressed by

$$
ThNi5 \xrightarrow[CO/H_2]{O_2} Ni/ThO_2,
$$

yielding a thoria-supported Ni catalyst. Compounds involving Fe, Co, and Ni with Al or Si (5, 6) also have been studied. Among the combinations investigated, the Th-Ni system seems to work best for the methanation reaction. Activity studies have shown (7, 8) that the relative methanation activities of catalysts derived from the alloys are $ThCo₅ < ThNi₅ > UNi₅ > ZrNi₅$.

Electron spectroscopy for chemical analysis (ESCA) $(8-12)$, and Auger electron

used to study a number of intermetallic alloy systems. However, no extensive investigation by a combination of several surface techniques has been reported for a single system of alloy catalysts. This type of study has proved to be valuable for studying conventional supported catalysts (15, 16). In the present work, we report the use of ESCA, AES, and ion scattering spectroscopy (ISS) to examine a series of Th_rNi_v intermetallic alloys $(Th_7Ni_3, ThNi, ThNi_2,$ $ThNi₅$). The investigation included surface characterization of fresh alloys and those treated with different reactive gases (O_2, Q_3) air, $CO/H₂$). Catalytic activity data for the oxidized intermetallic catalysts are given and compared with literature values (I) . A correlation between the results of surface characterization and catalytic activity is included.

spectroscopy (AES) (7, 13, 14) have been

EXPERIMENTAL SECTION

Alloys were prepared by melting a stoichiometric mixture of the metals in a watercooled copper boat under an atmosphere of purified argon as described by Imamura and Wallace (1). The intermetallics were then ground and sieved to 80 mesh. Studies of fresh intermetallics were accomplished immediately after grinding to avoid extensive air oxidation. Treatment of these alloys was carried out in a flow-type microreactor under air, O_2 and synthesis gas (H₂/CO = 3), respectively. When synthesis gas was used, the crushed catalysts were pretreated with Hz for 2 h to remove the oxide film. All treatments were carried out at 350°C. X-Ray diffraction analysis of the alloys was performed before and after treatment using a Diano XRD-6 diffractometer.

ESCA studies were carried out using an AEI ES-200A electron spectrometer equipped with an aluminum anode and an AEI DS-100 data system. The spectrometer was operated at 12 kV and 22 mA and the base pressure was below 10^{-8} Torr. The digital data obtained from the DS-100 data system were processed with an Apple II microcomputer. Binding energies were referenced to the Cls line at 284.6 eV. Sealable ESCA probes (16) were used to transfer air sensitive samples from the reaction chamber to the spectrometer.

A Physical-Electronics Model 545 Auger electron spectrometer was used to obtain Auger spectra of the alloys. The energy and emission current of the primary electron beam were set at 5 keV and 2 mA, respectively. The base pressure was below 10^{-8} Torr.

ISS studies were carried out using a 3M Model 525 ion scattering spectrometer equipped with a cylindrical mirror analyzer (CMA), to measure the energy of the elastically scattered ions. The instrument was backfilled with ²⁰Ne to 2×10^{-5} Torr after a base pressure below 5×10^{-9} Torr was obtained. The primary ion beam was operated at 2 keV and at a current density of 2×10^{-7} $A/cm²$.

Catalytic measurements were obtained for oxygen-treated intermetallics (350°C, 20 h). They were done in a microreactor operating at atmospheric pressure. Prior to each run, a sample mixture (50 mg catalyst $+$ 50 mg ground quartz) was reduced in H_2 at 275°C for 16 h. The catalyst mixture was then cooled to room temperature and chemisorbed with CO in a H_2 stream. After the amount of chemisorbed CO was measured, the catalyst was purged with H_2 and heated to 275°C. Hydrogen gas was then exchanged with a synthesis gas mixture having the composition 10% CO, 89% H₂, 1% Ar. Samples of the product were transferred to a gas chromatograph for detection and measurement.

RESULTS

Untreated Alloys

Surface characterization of the untreated alloys (Th₇Ni₃, ThNi, ThNi₂, ThNi₅) was done using ESCA and AES. The chemical states of the elements on the surfaces of the fresh alloys were determined by ESCA. Examination of the Ni $2p_{3/2}$, lines shown in Fig. 1, and the Th $4f_{7/2}$ spectra of the original intermetallic compounds indicates the presence of Ni, NiO (Ni $2p_{3/2} = 852.2$ and 854.5 eV), and ThO₂ (Th $4f_{7/2} = 333.6$ eV) as major surface species. Quantitative ESCA results were obtained using integrated peak areas for Ni $2p_{3/2}$ and Th $4f_{7/2}$. Estimates of the Ni surface concentrations for the alloys can be obtained from

$$
\frac{N_{\text{Ni}}}{N_{\text{Th}}} = \frac{I_{\text{Ni}}}{I_{\text{Th}}} \times \frac{\sigma_{\text{Th}}}{\sigma_{\text{Ni}}} \times \sqrt{\frac{E_{\text{Th}}}{E_{\text{Ni}}} \times \frac{D_{\text{Th}}}{D_{\text{Ni}}}}
$$
 (1)

using Scofield photoionization cross sections (17), where $N =$ number of atoms, $I =$ intensity (peak area), σ = Scofield photoionization cross section, $E =$ kinetic energy of photoelectron, $D =$ detector efficiency. D increases linearly with E in the AEI instrument (28). The Ni surface concentrations are plotted as a function of bulk Ni concentration in Fig. 2. It can be seen that all untreated alloys are Ni-poor on the surface. All points fall below the diagonal line. Nickel increases linearly with increasing bulk concentration and then drastically increases after 67% Ni. The ESCA results were confirmed by AES data. A plot of

FIG. 1. Ni $2p_{3/2}$ ESCA spectra for ThNi_s with different treatments. Th $O₂$ was observed in all cases. (A) Untreated. (B) In air (350°C). (C) In O₂ (350°C). (D) Either H₂ or CO/H₂ (350°C). (E) O_2 (350°C) followed by $H₂$ (400 $^{\circ}$ C).

AES (not shown) relative Ni intensity, Ni/ $Th + Ni$, calculated using peak to peak heights of the Ni $L_3M_{4,5}M_{4,5}$ and the Th $N₇O₅O₅$ lines shows exactly the same trend as the ESCA data.

Treated Alloys

Effect of reactive gases. The effect of reactive gases $(O_2, \text{air}, \text{CO/H}_2)$ on the alloys was studied in detail. As an example, reduction of ThNi_s in H_2 resulted in surface

segregation of Ni. The atomic Ni/Th ratio was found to increase from 2.0 for untreated ThNis to 8.8 when the alloy is reduced in H₂ at 350° C for 2 h. Further surface enrichment of Ni was observed when the catalyst was treated with synthesis gas for longer periods $(6 h)$ (Ni/Th = 12.5). The surface of this reacted catalyst consists primarily of metallic nickel (Ni $2p_{3/2} = 852.2$ eV) and ThO₂ (Th $4f_{7/2} = 333.6$ eV) (Fig. 1). X-Ray diffraction of the same sample also indicated decomposition of the bulk intermetallic into Ni and $ThO₂$.

Table 1 shows the effect of O_2 on the surface atomic ratios of ThNi₅. The atomic Ni/ Th ratio decreased from 2.0 to 1.4 when the sample was oxidized in air for 20 min at 100°C. Longer oxidation times (lOO"C, 1 h) resulted in an even smaller atomic Ni/Th ratio (0.97). More drastic conditions $(200^{\circ}C, 20 \text{ min})$ caused the bulk intermetallic to decompose into Ni and $ThO₂$ as observed by X-ray diffraction; under these conditions the atomic Ni/Th ratio began to increase (Ni/Th $= 2.9$). Higher temperature oxidation (35O"C, 1 h) resulted in further decomposition of the intermetallic as well as a further increase in the atomic Ni/Th ratio (Ni/Th = 6.0). Although ThO₂ (Th $4f_{7/2}$) $= 333.6$ eV) was observed by ESCA for all treatments, the surface Ni was completely

FIG. 2. Relative Ni surface concentration of Th-Ni alloys. (\blacksquare) Before treatment. (\blacktriangle) After treatment. The diagonal line indicates equal surface and bulk concentrations.

| Treatment | ESCA binding energies," eV | | Atomic ratio ^b | Phase(s) (XRD) | |
|---|----------------------------|--------------|---------------------------|---|--|
| | Ni $2p_{3/2}$ | Th $4f_{12}$ | Ni/Th | | |
| Untreated | 852.2, 854.5 | 333.6 | 2.0 | ThNi | |
| In air at 100° C for 20 min | 852.2, 854.9 | 333.6 | 1.4 | ThNi, | |
| In air at 100° C for 60 min | 852.2, 854.7 | 333.6 | 0.97 | ThNi | |
| In air at 200° C for 20 min | 852.2, 854.4 | 333.6 | 2.9 | ThNi _s , Ni, ThO ₂ | |
| In air at 350° C for $1h$ | 853.8, 855.6, 861.0 | 333.6 | 6.0 | ThNi _s , Ni, ThO ₂ , NiO | |
| In air at 350° C for $8h$ | 853.8, 855.6, 861.0 | 333.6 | 13.2 | Ni. ThO ₂ , NiO | |

Atomic Ratios of ThNi_s for Various Treatments With Air

 α Binding energies are measured with a precision of ± 0.15 eV.

 \bar{b} Atomic ratios are calculated from ESCA data using Scofield photoionization cross section (17). Data are measured to $\pm 6\%$ RSD.

oxidized to NiO (Table 1) only at high temperature (3SO°C, 1 h).

Segregation of Ni to the surface also changed with length of treatment time. Fig. 3 shows a plot of the atomic ratio (Ni/Th) versus oxidation time for ThNi_s. A drastic increase in the Ni signal was observed during the first 4 h of oxidation. Further treatment resulted in a small gradual change, reaching a steady state value after about 8 h of oxidation.

The extent of reduction of oxidized ThNi_s was also studied using the "ESCA- H_2 titration" technique (15, 16). The oxi-

FIG. 3. Change of the Ni/Th atomic ratio with oxidation time for ThNi, at 350°C.

dized catalysts were reduced in flowing H_2 at 400°C for 4 h and then examined by ESCA without exposure to air. ESCA data showed that surface Ni was completely reduced, as indicated by the appearance of only metallic nickel (Ni $2p_{3/2} = 852.2$ eV) (Fig. 1).

The effect of oxidation time on other alloys (Th₇Ni₃, ThNi, ThNi₂) was also studied. Surface concentrations of Ni for all alloys (Th₇Ni₃, ThNi, ThNi₂) were unchanged after 8 h of oxidation. As it can be seen in Table 2, there are no significant differences in atomic Ni/Th ratios whether the alloys were treated for 8 or 20 h. X-Ray diffraction data showed that all alloys were decomposed completely into Ni and $ThO₂$ after 8 h of oxidation.

Comparison of the effects of different reactive gases on the alloys is shown in Table 2. The results for ThNi₅ in Table 2 for reaction with air, O_2 , or CO/H_2 show that the Ni/Th atomic ratios are the same for all three treatments within experimental error $(\pm 6\%$ RSD). The same effect was observed for the other alloys: Th_7Ni_3 , ThNi, and ThNi₂. Thus, treatment with different gases results in the same degree of Ni surface segregation. in addition, Table 2 also lists the

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| Intermetallic | Treatment $(350^{\circ}C)$ | Atomic ratio a Ni/Th | Surface species | Bulk phase $(XRD)^b$ |
|-------------------|--|---------------------------|---------------------------|---------------------------|
| Th_7Ni_3 | Untreated | 0.024 | Ni, NiO, ThO ₂ | Th_7Ni_3 |
| | In CO/H ₂ for 6 h^b | 0.026 | Ni. ThO ₂ | Ni, ThO ₂ |
| | In $O2$, for $8h$ | 0.027 | NiO, ThO ₂ | Ni, NiO, ThO, |
| | In $O2$ for 20 h | 0.026 | | |
| ThNi | Untreated | 0.15 | Ni, NiO, ThO, | ThNi |
| | In H ₂ /CO for 6 h ^b | 0.24 | Ni. ThO ₂ | Ni , ThO , |
| | In O_2 for $8 h$ | 0.26 | NiO, ThO ₂ | Ni, NiO, ThO ₂ |
| | In $O2$, for 20 h | 0.26 | | |
| ThNi ₂ | Untreated | 0.26 | Ni, NiO, ThO ₂ | ThNi ₂ |
| | In H ₂ /CO for 6 h ^b | 0.42 | Ni, ThO, | Ni , Th $O2$ |
| | In O_2 for $8 h$ | 0.44 | NiO. ThO ₂ | Ni, NiO, ThO ₂ |
| | In $O2$, for 20 h | 0.47 | | |
| ThNi _s | Untreated | 2.0 | Ni, NiO, ThO ₂ | ThNi, |
| | In H , for 2 h | 8.8 | Ni , Th $O2$ | ThNi |
| | In H ₂ /CO for 6 h ^b | 12.5 | Ni. ThO ₂ | Ni, ThO, |
| | In air for 8 h | 13.2 | NiO, ThO ₂ | Ni, NiO, ThO ₂ |
| | In O_2 for 8 h | 12.5 | NiO, ThO ₂ | Ni, NiO, ThO ₂ |
| | In $O2$ for 20 h | 13.2 | | |

Atomic Ratios of Thorium-Nickel Alloys

 α Atomic ratios are calculated from ESCA data using Scofield photoionization cross section (17). Data are measured to ± 6 RSD.

 Φ All synthesis gas treatments are prereduced in H₂ at 350°C for 2 h.

species found on surfaces of the reacted alloys by ESCA. Ni and $ThO₂$ were found in alloys decomposed by synthesis gas while NiO and $ThO₂$ existed on oxidized alloys.

Relationship between relative Ni intensity and bulk concentration. The relationship between relative Ni intensity (Ni/Th $+$ Ni) of the oxygen-reacted catalysts versus bulk Ni content was examined by both ESCA and AES. In the case of ESCA, the relative Ni intensity can be corrected to represent the relative surface concentration of Ni as shown in Fig. 2. Both ESCA and AES show segregation of Ni to the surface when the alloys are treated. Note the unusual enrichment of Ni on the surface of ThNis relative to the other Th-Ni alloys which are still Ni poor with respect to the bulk (Fig. 2).

Figure 4 shows ISS spectra of the oxygen-reacted alloys. The Ni signal is weak for Th_7Ni_3 , ThNi, and ThNi₂, but it is quite significant for ThNi₅. A plot of ISS Ni relative intensity ($Ni/Th + Ni$) versus bulk Ni concentration, shown in Fig. 5, indicates constant relative Ni intensity with increasing bulk Ni content for the first three re-

FIG. 4. ISS spectra for oxygen-treated Th-Ni alloys.

FIG. 5. Relative intensity ratio (Ni/Ni + Th) for oxygen-treated Th-Ni alloys. (H) ISS. (A) Valence AES.

acted intermetallics, but a significant difference for ThNi_s.

Information about the outermost atomic layers of the treated alloys can also be obtained from Auger lines involving valenceelectron transitions. There is some overlap between the valence peaks of TH, O_4P_2V $(65 eV)$, and Ni, MVV $61 eV$), as shown in Fig. 6. Although peak-to-peak heights of the Ni MVV peak (61 eV) in all four alloys are not much affected by the Th O_4P_2V (65 eV) peak, the converse is not true. The Th O_4P_2V (65 eV) lines of ThNi_s completely overlap the Ni MVV signal (61 eV). Thus, the Th O_5 VV peak (81 eV) was used for analysis. A plot of relative Ni intensity, Ni/ Th + Ni, using peak-to-peak height of Ni MVV (61 eV) and Th O_5 VV (81 eV) versus bulk Ni content (Fig. 5) shows a trend similar to that observed for the ISS data (Fig. 5).

Methanation Reaction

Activity data for oxygen-treated catalysts are summarized in Table 3. Turnover numbers and rates of CO conversion to hydrocarbon are given at 3 and 7 h. All catalysts showed a slight decrease in activity with time on stream. Although ThNi_s was found to be the most active catalyst, rates corrected per gram of Ni decreased rapidly with increasing Ni content (Table 3). The

same phenomenon was observed for sorption capacity. As shown in Table 3, CO chemisorption of ThNis is larger than that of any other Th-Ni catalyst. However, the metal dispersion derived from sorption data showed a rapid decline with increasing Ni content (Table 3). The calculation was made based on the assumption that one CO molecule interacts with one Ni metal atom (I). Methane and ethane were the two main products in CO hydrogenation of Th-Ni catalysts. Their fractions are listed in Table 3. Methane selectivity changes with Ni content. An alloy with higher Ni content generally gives a larger fraction of methane. It is noteworthy that the methane selectivity also increased with time on stream.

The relative Ni surface concentrations of the used catalysts estimated from ESCA data are listed in Table 3. Values for oxidized alloy catalysts are also given for comparison. The relative surface concentrations of Ni in used catalysts were found to be smaller than those of oxidized catalysts,

FIG. 6. Valence AES spectra of oxygen-treated Th-Ni alloys.

TABLE 3

Characterization and Activity of Th-Ni Catalysts"

| Alloy | $Th_2Ni_2^b$ | ThNi ^b | ThNi ₂ | ThNi. |
|--|----------------------|----------------------|----------------------|----------------------|
| CO sorption $(\mu mol/g)$ | 42 | 80 | 66 | 98 |
| Overall dispersion $(\%)$ | 2.9 | 2.7 | 1.3 | 1.2 |
| CO conversion rate $(cm3/g/sec)$ | | | | |
| 3 _h | 5.8×10^{-2} | 7.2×10^{-2} | 5.2×20^{-2} | 12×10^{-2} |
| 7 _h | 4.9×10^{-2} | 5.1×10^{-2} | 3.6×10^{-2} | 9.6×10^{-2} |
| Normalized rate ^d (cm ³ /g Ni/sec) | | | | |
| 3 _h | 0.68(0.67) | 0.42(0.40) | 0.18(0.17) | 0.26(0.23) |
| 7 _h | 0.57(0.56) | 0.30(0.28) | 0.13(0.12) | 0.21(0.18) |
| Turnover number (\sec^{-1}) | | | | |
| 3 _h | 0.061 | 0.042 | 0.035 | 0.054 |
| 7 _h | 0.051 | 0.029 | 0.025 | 0.044 |
| Fraction of methane formed $(\%)$ | | | | |
| 3 _h | 83 | 85 | 85 | 86 |
| 7 _h | 85 | 86 | 87 | 87 |
| Fraction of ethane formed $(\%)$ | | | | |
| 3 _h | 17 | 15 | 15 | 14 |
| 7 _h | 15 | 14 | 13 | 13 |
| Relative surface concentration of Ni $(\%)$ | | | | |
| Oxidized | 2.5 | 21 | 32 | 96 |
| Used | 1.3 | 12 | 15 | 87 |

^{*a*} H₂/CO = 9, Rates obtained at 275^oC.

 b Treated in O₂ for 20 h at 350°C.

 ϵ Treated in O₂ for 16 h at 350°C. Temperature was raised slowly to avoid sintering.

^d Assume decomposition products are NiO/ThO₂ and Ni/ThO₂ for values in parentheses.

which suggests that some sintering has occurred in the used catalysts.

DISCUSSION

Untreated Alloys

Magnetic (II) and ESCA (9) investigations of LaNi₅ have shown that exposure of LaNis to low levels of $O₂$ results in surface segregation of La to form $La₂O₃$, along with precipitation of Ni to form superparamagnetic metallic Ni particles. Surface segregation of Th as well as oxidation of Th and Ni have been observed by Schlapbach and Brundle (9) in an ESCA study of ThNi_s when this alloy is oxidized mildly (10^9) L $O₂$). Transformations similar to those of LaNi_s would be expected for ThNi_s because of similarities between the two systems.

In the present study, the untreated alloys have been oxidized by exposure to air at room temperature, which corresponds to

more severe oxidizing conditions than those used by Schlapbach and Brundle (9). Thus, the surface transformation of the alloys discussed above and the consequences, such as an Ni-poor surface with respect to the bulk and the existence of $ThO₂$, Ni, and NiO as surface species, were observed.

Treated Alloys

The Th-Ni alloys were decomposed by all of the reactive gases studied: air, $O₂$, and $CO/H₂$. In the cases of $O₂$ and air, it is suggested that (1) oxygen reacts selectively with thorium to form $ThO₂$ due to the strong affinity of thorium for oxygen (ΔH_f) $= -292$ kcal/g mol). This results in precipitation of Ni from the original intermetallic, which will then segregate to the surface. An analogous mechanism should be expected for CO since it has been shown (19) that CO is dissociated into C and O in the methanation reaction. This dissociated 0 should affect the Th-Ni alloys in a similar manner to regular oxygen.

Comparison of relative Ni surface concentrations of all alloys under different gas treatments has shown that they are the same within experimental error $(\pm 6 \text{ RSD})$. Since Ni is the catalytically active species in the methanation reaction, activities of the alloy catalysts are expected to be similar whether they are formed by treatment with air, O_2 , or CO/H₂. Indeed, the activity data for the methanation reaction of oxidized ThNi_s and ThNi_s decomposed by synthesis gas, show that they are not very different (1). Similar activities were also observed for Ni-Si and Co-Si (5) alloys, decomposed by either O_2 or CO/H₂.

The extent of Ni reduction studied by the $ESCA-H₂$ titration technique has been valuable for detecting the presence of metal-support interactions in conventional supported catalysts. For example, in γ -alumina-supported catalysts (15) only part of the surface Ni is reducible. This can be explained by formation of two Ni species on the surface: "readily reduced Ni" and "hard to reduce Ni," which have been assigned as NiO and a "surface aluminate," respectively. Results from the $ESCA-H₂$ titration of oxidized ThNi₅ show that all of the surface Ni is reduced by H_2 (Fig. 1D). The existence of both NiO (Ni $2p_{3/2} = 853.8$) and 855.6 eV) on oxidized ThNis and the facile reducibility of the Ni species (Fig. 1). indicate that there is essentially no interaction between the transition metal and the support in Th_xNi_y intermetallic catalysts. It is noteworthy that $ESCA-H₂$ titration measurements have also been performed on Ni/ ThO₂-supported catalysts (20) prepared by conventional impregnation methods. No metal-support interaction was observed for the conventional thoria-supported Ni catalysts.

Relationship between relative Ni intensity and bulk concentration. Examination of the relative intensity of an element versus its bulk concentration has been valuable

FIG. 7. Change in concentration of thorium-nickel alloys. D_1 = Difference between bulk and surface concentration of Ni fresh alloys $[(N_{\text{Ni}}/N_{\text{Ni}} + N_{\text{Th}})_{\text{bulk}}$ - $(N_{\rm Ni}/N_{\rm Ni} + N_{\rm Th})_{\rm surface}$].

for detecting changes in surface properties of alloy systems as the composition is changed (21) . Such differences are usually indicated by observing changes in the slope of a plot of relative intensity vs bulk composition. Plots of both ESCA (Fig. 2) and AES relative Ni intensities of fresh alloys versus bulk Ni concentration show a discontinuity at 67% Ni. This suggests a surface property change when the Ni concentration is increased beyond 67%.

As was discussed in previous sections, exposure of the fresh alloys to air caused transformation of the alloy surface to Ni, $NiO/ThO₂$ along with the segregation of thorium to the surface, which subsequently results in an Ni poor surface relative to the bulk. The degree of transformation depends on temperature and the stability of the alloy. At room temperature, a less stable alloy would undergo surface transformation more readily and surface segregation of thorium more extensively. As a consequence, a smaller Ni surface concentration compared to the bulk is obtained, which gives rise to a larger difference between bulk and surface concentrations of Ni. Thus, a plot of Ni concentration difference between the bulk and surface of the fresh alloys: $D_1(D_1 = N_{\text{Ni}}/(N_{\text{Ni}} + N_{\text{Th}})_{\text{bulk}} - N_{\text{Ni}}/N_{\text{H}}$ $(N_{\text{Ni}} + N_{\text{Th}})_{\text{surface}}$, versus bulk Ni content (Fig. 7) would give information regarding the stability of the alloys. As shown in Fig. 7, D_1 increases linearly with bulk Ni content and then drastically decreases for ThNi₅. This suggests that except for ThNi₅ which is unusually stable, an alloy with larger Ni content would be less stable. It is also noteworthy that ThNi₂ and ThNi are found to be less stable than Th_7Ni_3 and ThNi₅. This has also been reported by Imamura and Wallace (1) who observed that ThN_i, and ThN_i are readily decomposed at room temperature while Th_7Ni_3 and $ThNi_5$ are decomposed completely only at 350°C.

ESCA (22) and AES (23) generally give information about 10–20 atomic layers while ISS (24) is considered to be sensitive to the first one or two atomic layers. Examination of ESCA (Fig. 2) and AES data for treated alloys (not shown) indicates that for the top IO-20 atomic layers, Ni concentration increases with bulk Ni content and an unusual accumulation of Ni relative to the other Th-Ni alloys is observed for ThNi₅. It is noteworthy that except for $ThNi₅$ the surfaces of even the reacted catalysts are still Ni poor with respect to the bulk (Table 2). The large enrichment of Ni on the surface of treated ThNi_s is again shown in ISS results (Fig. 5). However, Ni concentrations in the first atomic layers of the other three treated alloys are essentially unchanged and are significantly lower than Th $Ni₅$. ISS results are also confirmed by valence AES data as shown in Fig. 5.

Methanation Reaction

CO chemisorption and activity data for the methanation reaction of Th_xNi_y catalysts are in general agreement with those reported by Imamura and Wallace (I). Similar variations in CO sorption and activity with bulk Ni content were obtained. ThNis was the most active catalyst while the other three alloys Th_7Ni_3 , ThNi, and ThNi₂ showed comparable activity with each other (Table 3).

Comparison between our results for Th,Ni, intermetallic catalysts and those of 5% Ni supported on alumina has shown that only ThN $i₅$ is more active than the alumina-supported catalyst. CO chemisorption and turnover number for total conversion at 275 °C of 5% $Ni/Al₂O₃$ were found to be 110 μ mol/g and 0.038/sec, respectively. The corresponding rate is 9.5×10^{-2} cm³/g/ sec. ThNis showed lower sorption capacity $(97 \mu \text{mol/g})$ but a higher CO conversion rate $(12 \times 10^{-2} \text{ cm}^3/\text{g/sec})$ and turnover number $(0.054 \text{ sec}^{-1}).$

Surface Characterization and Methanation Activity of Th_xNi_y Alloys

Comparison of the surface analysis and activity (Table 3) shows that there is some interdependence between the surface concentration of Ni and activity; high Ni surface concentration generally results in greater methanation activity. For example, the high activity of ThNi_s relative to other Th-Ni alloys can be explained by the large enrichment of surface Ni (Table 3). The similar activities of Th_7Ni_3 , ThNi, and Th $Ni₂$ can be accounted for by the equivalent Ni intensity of those three alloys detected by ISS and valence band AES. However, a quantitative correlation between activity and surface concentration of Ni in $Th₇Ni₃$ and ThNis has shown that while the activity of ThNi_s is twice that of Th₇Ni₃, the relative surface concentration of ThNis is 30 times larger than that of Th_7Ni_3 . This inconsistency can possibly be explained by the presence of different types of active sites. It is possible that thorium could play an active role in the methanation reaction in addition to its function as a dispersing agent. Thorium has been used as a promoter in several types of reactions. Since thorium content in Th_xNi_y alloys decreases with increasing Ni concentration, the amount of Th which is available for incorporation with Ni in ThNi_s is very much less than that of Th_7Ni_3 . Thus, there are possibly two types of Ni in Th $Ni₅$. The active Ni which is tightly interacted with thorium and the less active one which is surrounded by Ni itself. Since surface techniques (ESCA, AES, ISS) cannot differentiate different

FIG. 8. Models for thorium-nickel catalysts. (\bullet) Th or $ThO₂$. (O) Ni or NiO.

types of Ni species, the large surface concentration of Ni in ThNis can be misleading.

The lack of correlation between Ni surface concentration and the activities of Th_7Ni_3 and ThNi_s also can be interpreted by considering that some Ni in Th_7Ni_3 is covered by $ThO₂$ and therefore cannot be detected by ESCA. This "unseen Ni," however, must chemisorb CO and be reactive (Table 3). It is suggested that some type of aggregate is formed as a result of oxidation; a possible type of aggregate is shown in Fig. 8. The clusters are composed of $ThO₂$ which forms a cover and Ni which fills the pores. Any excess Ni will be on top of the ThO₂ cover. This cover of $ThO₂$ is probably spongy and can be penetrated by hydrogen as well as CO. Thus, Ni under this permeable ThO₂ layer is readily reacted and can chemisorb CO.

In the case of Th $Ni₅$, the same model can apply. Since $ThNi₅$ is a Ni-rich compound, it is likely that Ni will be deposited on top of $ThO₂$ spongy layers. This effect will result in a large accumulation of Ni on the surface. Models for ThNi and ThNi₂ should resemble both Th_7Ni_3 and ThNi₅, but should be more analogous to Th_7Ni_3 because the Ni concentrations at the surfaces of ThNi and ThNi, are smaller than those of the bulk and they are equivalent to that of Th_7Ni_3 (Fig. 5). Although our data do not permit exact distinction between the two models suggested, we believe that the second (as shown in Fig. 8) is the more likely.

SUMMARY

1. Untreated alloys have already undergone surface transformation into Ni, NiO, and $ThO₂$. All are Ni-poor on the surface with respect to the bulk.

2. Treatment of the alloys with $CO/H₂$, air, or $O₂$ results in both surface and bulk decomposition of the original intermetallic into either $Ni/ThO₂$ or NiO/ThO₂.

3. In all cases, Ni is observed to segregate to the surface when the intermetallics are treated. The degree of segregation is found to be the same for different reactive gases.

4. Catalysts with high surface concentrations of Ni generally show greater methanation activity. Lack of a quantitative correlation between the activity and surface Ni concentration of Th_7Ni_3 and ThNi_s is probably due to different types of Ni -ThO₂ aggregates being formed in the Ni-poor (Th_7Ni_3) and Ni-rich $(ThNi₅)$ alloys.

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